Synthesis and Crystal Structure of σ -Zn_{0.25}V₂O₅ · H₂O with a Novel Type of V₂O₅ Layer¹

Yoshio Oka,* Osamu Tamada,† Takeshi Yao,‡ and Naoichi Yamamoto†

*Department of Natural Environment Sciences, Faculty of Integrated Human Studies; †Graduate School of Human and Environmental Studies; and ‡Division of Energy and Hydrocarbon Chemistry, Graduate School of Engineering, Kyoto University, Kyoto 606, Japan

Received November 28, 1995; in revised form June 13, 1996; accepted June 17, 1996

A new layered vanadium bronze designated as σ -phase $Zn_{0.25}V_2O_5 \cdot H_2O$ has been hydrothermally synthesized from ZnCl₂ and VO(OH)₂. A single-crystal study revealed the triclinic system P1: a = 10.614(2) Å, b = 8.031(3) Å, c = 10.7688(9)Å, $\alpha = 90.65(1)^{\circ}$, $\beta = 91.14(1)^{\circ}$, $\gamma = 90.09(2)^{\circ}$, and Z = 8. The structure was solved and refined to $R/R_{\rm w} = 0.079/0.048$ for 1580 reflections with $I > 3\sigma(I)$, which consists of V₂O₅ layers stacking along the c axis and interstitial hydrated Zn^{2+} ions. The V₂O₅ layer adopts a novel polyhedral framework built up of VO₆ octahedra, VO₅ trigonal bipyramids, and VO₄ tetrahedra. When projected parallel to the *ab* plane, the V_2O_5 layer is described by the atomic sheet model in a manner similar to the V_2O_5 layer of δ phase such as δ -Ag_x V_2O_5 , that is, a doublesheet type composed of two V₂O₅ sheets facing each other. The interstitial Zn atom forms a ZnO_6 octahedron with two apical oxygens of the VO₄ tetrahedra on opposide sides and with four coplanar water molecules. The anhydrous σ phase was obtained by heating the hydrate up to 200°C accompanied by the contraction in layer spacing from 10.76 to 8.92 Å. © 1996 Academic Press, Inc.

INTRODUCTION

Vanadium (IV, V) oxide bronzes $M_x V_n O_m$ often adopt layered structures consisting of $V_n O_m$ layers and interlayer M ions that are sometimes in hydrous states. $V_n O_m$ layers so far known have compositions of V_2O_5 , V_3O_8 , and V_4O_{11} of which a major group of the $M_x V_2O_5$ family has been extensively studied (1, 2). In the $M_x V_2O_5$ family there are two primary phases α - and δ - $M_x V_2O_5$, where to avoid complexity δ - $M_x V_2O_5$ represents the related phases named

 ε -, ρ -, and ν - M_x V₂O₅ as well. The V₂O₅ layers of the α and δ phases are structurally related to the oxide V₂O₅ (3) and $VO_2(B)$ (4), respectively, and are also described by atomic sheet models as a single-sheet type for α phase and a double-sheet type for δ phase (5). Hagenmuller *et al.* (6, 7) studied α phases for a number of M species and found rather narrow ranges of x up to usually 0.03 and at most 0.11. They also reported an α' phase for M = Na with x = 0.7 to 1.0 that is basically isomorphous to α phase (8). The structure of δ phase was first determined by Anderson (9) for Ag_{0.68}V₂O₅, and recently δ and related phases having double-sheet-type V₂O₅ layers were sorted and reviewed by Galy (10). Hydrated δ phases of $M_x V_2 O_5 \cdot nH_2 O_5$ were hydrothermally synthesized and structurally characterized by the present authors for M = alkali metal ions (11, 12) and for $M = \text{VO}^{2+}$ (13). The values of x for δ phase range more widely from 0.1 for $M = VO^{2+}$ (13) to 1.0 for $M = Cu^+$ (14). Besides the two phases γ -LiV₂O₅ has another type of V_2O_5 layer which may be grouped into a single-sheet type (15). It is of interest to add a new type of layer structure in the $M_x V_2 O_5$ family. By using a hydrothermal method we have succeeded in synthesizing a new member of the $M_x V_2 O_5$ family for M = Zn formulated by a hydrous form $Zn_{0.25}V_2O_5 \cdot H_2O$. This new phase is named here as σ phase. A single-crystal study revealed a novel structure of V_2O_5 layer with a double-sheet type. Comparisons of δ phase with δ phase for Zn compounds are also made.

EXPERIMENTAL

Sample Preparation

Hydrothermal synthesis was carried out in a $ZnCl_2-VO(OH)_2$ system, where $VO(OH)_2$ powders were prepared in advance hydrothermally at 150°C from a mixture of $VOSO_4$ and NaOH. A suspension of $VO(OH)_2$ powders in a 0.1 *M* ZnCl₂ aqueous solution was sealed in a Pyrex ampoule and was treated in an autoclave at 280°C for 40 h. Products of shiny black powders were separated by filtra-

¹ See NAPS document No. 05329 for 54 pages of supplementary material. Order from ASIS/NAPS, Microfiche Publications, P.O. Box 3513, Grand Central Station, New York, NY 10163. Remit in advance \$4.00 for microfiche copy or for photocopy, \$7.75 for up to 20 pages plus \$.30 for each additional page. All orders must be prepaid. Institutions and organizations may order by purchase order. However, there is a billing and handling charge for this service of \$15. Foreign orders add \$4.50 for postage and handling, for the first 20 pages, and \$1.00 for each additional 10 pages of material, \$1.50 for postage of any microfiche orders.



FIG. 1. Scanning electron micrograph of hydrothermally synthesized σ -Zn_{0.25}V₂O₅·H₂O powders.

tion and washed thoroughly with distilled water. Powder X-ray diffractometry showed a highly oriented layered phase sometimes with small amounts of impurity phases of VO₂(A) and/or rutile-type VO₂. A chemical composition of the layered phase was determined to be $Zn_{0.25(1)}V_2O_5 \cdot nH_2O$ ($n \approx 1.0$) by an energy dispersive X-ray analysis, an atomic absorption analysis and a thermogravimetric analysis. As shown by the SEM picture in Fig. 1 the layered phase exhibits a flat rhombic shape. High-temperature X-ray diffraction measurements were carried out by using a Rigaku Rad-B system with monochromated CuK α radiation where a sample was heated by blowing hot air onto the surface to control temperature within $\pm 1^{\circ}C$.

Single-Crystal X-Ray Diffraction

Single-crystal X-ray studies were performed on a crystal of dimensions $0.30 \times 0.08 \times 0.01$ mm. A preliminary study was made by using a Rigaku AFC-5S diffractometer and final data collection was made by using a Rigaku AFC-7R with higher intensity MoK α radiation. The triclinic system was detected with lattice parameters of a = 10.614(2) Å, b = 8.031(3) Å, c = 10.7688(9) Å, $\alpha = 90.65(1)^\circ$, $\beta =$ $91.14(1)^\circ$, and $\gamma = 90.09(1)^\circ$ determined from 24 reflections in $20.4^\circ \le 2\theta \le 27.4^\circ$. Intensity data were collected up to $2\theta = 80^\circ$ together with standard reflections of 5 3 0, -530, and 5-30 which were monitored every 150 reflections to give no significant intensity fluctuation. An empirical correction of absorption effect was made using the ψ scan method, resulting in transmission factors of 0.731– 1.000. A total of 8734 unique reflections with I > 0 were obtained of which 1580 reflections with $I > 3\sigma(I)$ were used in the structure refinement. No more than 18% of total reflections were effective because of low signal-to-noise ratios of intensity profiles due to poor quality, thin shape, and tiny size of the crystal.

Structure Determination

The structure determination was performed by the Patterson and Fourier method using the program FRAXY (16). An initial model was derived based on the centrosymmetric space group $P\overline{1}$ and Z = 8. Preliminary structure refinements were executed using the program RADY (17). Atomic scattering factors for neutral atoms were taken from the "International Tables for X-ray Crystallography IV" (18). Sites of V atoms were successfully determined, which helped to locate oxygens and interlayer Zn atoms. Oxygens of interlayer water molecules denoted Ow were subsequently located in differential Fourier maps. The occupancies of interlayer Zn and Ow sites as well as V sites were checked to give no appreciable deviation from full occupancy and thus they were fixed to unity, yielding a stoichiometric composition $Zn_{0.25}V_2O_5 \cdot H_2O$. It is noted that isotropic temperature factors were solely used throughout the refinements owing to poor quality of the crystal and to a limited number of data for calculating the variables. The full-matrix least-square refinements led to $R/R_{\rm w} = 0.079/0.048$ for 132 variables when the TEXSAN crystallographic software package (19) was used. The crys-

TABLE 1Crystallographic Data and ExperimentalParameters for σ -Zn_{0.25}V₂O₅ · H₂O

Chemical formula	$Zn_{0.25}V_2O_5 \cdot H_2O$
Space group	$P\overline{1}$
a (Å)	10.614(2)
b (Å)	8.031(3)
c (Å)	10.7688(9)
α (°)	90.65(1)
β (°)	91.14(1)
γ (°)	90.09(2)
Z	8
D_c/gcm^{-3}	3.130
Crystal size (mm)	0.30 imes 0.08 imes 0.01
Radiation	$MoK\alpha$
Scan technique	$2\theta - \omega$
Scan width, $\Delta \omega$ (°)	$1.73 + 0.3 \tan\theta$
Scan speed 2θ (° min ⁻¹)	16
$2\theta_{\rm max}$ (°)	80
No. unique reflections $(I > 0)$	8734
No. reflections $(I > 3\sigma(I))$	1580
No. variables	132
R	0.079
R _w	0.048

tallographic data and experimental parameters are listed in Table 1 and atomic parameters and isotropic temperature factors in Table 2.

RESULTS AND DISCUSSION

Description of Crystal Structure

As depicted in Fig. 2, a layered structure having a novel type of V_2O_5 layer has been disclosed which consists of V–O polyhedra of three kinds, namely, VO_6 octahedra for V(1), V(2), V(5), and V(6); VO₅ trigonal bipyramids for V(3) and V(7); and VO_4 tetrahedra for V(4) and V(8). We designate this novel phase as σ phase or more specifically σ -Zn_{0.25}V₂O₅ · H₂O to differentiate from existing phases of, e.g., α and δ . The bond distances and angles of the V–O polyhedra are listed in Table 3. It is indicated by the analysis of intensity data that the triclinic system can be approximated by the monoclinic system $P2_1/a$ when α and γ parameters are set at 90°. Actually the extinction rule for $P2_1/a$ did not completely hold; some reflections of 0k0with k = 2n + 1 or h0l with h = 2n + 1, for example, 010, 070, 100, and 102, showed weak but appreciable intensities. The triclinic system of this crystal is therefore said to be pseudo-monoclinic. In the monoclinic $P2_1/a$, atomic sites of (x, y, z) and (x + 1/2, 1/2 - y, z) are equivalent. This relationship is found to hold closely in following atomic pairs of the V_2O_5 layer: V(1)–V(5), V(2)–V(6), V(3)–V(7), V(4)-V(8), O(1)-O(7), O(2)-O(8), O(3)-O(9), O(4)-O(10), O(5)-O(11), O(6)-O(12), O(13)-O(17), O(14)-

TABLE 2Atomic Parameters and Isotropic Temperature Factors for σ -Zn_{0.25}V₂O₅ · H₂O

Atom	x	у	z	$B_{\rm eq}$ (Å ²)
V(1)	0.7014(5)	0.7187(6)	0.1412(5)	0.67(9)
V(2)	0.6876(4)	0.3407(6)	0.1418(5)	0.61(9)
V(3)	0.9159(4)	0.0412(6)	0.1381(5)	0.47(8)
V(4)	0.4749(5)	0.0340(6)	0.1628(5)	0.43(8)
V(5)	0.2012(5)	0.7859(6)	0.1407(5)	0.64(9)
V(6)	0.1879(4)	0.1623(6)	0.1403(5)	0.59(8)
V(7)	0.4161(4)	0.4632(6)	0.1403(5)	0.47(8)
V(8)	0.9746(14)	0.4701(6)	0.1612(5)	0.46(8)
Zn(1)	0	1/2	1/2	1.51(8)
Zn(2)	1/2	0	1/2	1.48(8)
O(1)	0.814(2)	0.522(2)	0.124(2)	0.3(3)
O(2)	0.581(2)	0.540(2)	0.110(2)	0.8(3)
O(3)	0.790(2)	0.194(2)	0.080(2)	1.0(3)
O(4)	0.526(1)	0.240(2)	0.124(1)	0.3(2)
O(5)	0.831(2)	0.868(2)	0.087(2)	1.0(3)
O(6)	0.571(1)	0.894(2)	0.097(2)	0.7(3)
O(7)	0.311(2)	0.986(2)	0.124(2)	0.3(3)
O(8)	0.082(2)	0.965(2)	0.104(2)	0.7(3)
O(9)	0.289(2)	0.310(2)	0.078(2)	0.9(3)
O(10)	0.026(1)	0.263(2)	0.120(2)	1.0(3)
O(11)	0.332(2)	0.635(2)	0.089(2)	1.1(3)
O(12)	0.071(2)	0.609(2)	0.099(2)	1.1(3)
O(13)	0.706(2)	0.738(2)	0.289(2)	1.4(4)
O(14)	0.700(2)	0.320(2)	0.283(2)	2.0(4)
O(15)	0.904(2)	0.042(2)	0.286(2)	1.5(4)
O(16)	0.488(2)	0.021(2)	0.312(2)	0.9(3)
O(17)	0.205(2)	0.768(2)	0.290(2)	1.6(4)
O(18)	0.202(2)	0.189(2)	0.285(2)	1.4(4)
O(19)	0.404(2)	0.465(2)	0.286(2)	1.0(3)
O(20)	0.989(2)	0.488(2)	0.309(2)	1.2(4)
Ow(1)	0.032(2)	0.759(3)	0.495(2)	3.5(5)
Ow(2)	0.802(2)	0.538(3)	0.510(2)	2.4(4)
Ow(3)	0.516(3)	0.264(3)	0.507(3)	4.4(6)
Ow(4)	0.303(2)	0.022(2)	0.507(2)	2.2(4)

O(18), O(15)–O(19), and O(16)–O(20). For simplicity we describe the structure of V_2O_5 layer based on $P2_1/a$ by using the first atoms of each pair. The V₂O₅ layer now consists of $V(1)O_6$ and $V(2)O_6$ octahedra, $V(3)O_5$ trigonal bipyramid, and V(4)O₄ tetrahedron; V(1)O₆, V(2)O₆, $V(3)O_5$, and $V(4)O_4$ represent $V(5)O_6$, $V(6)O_6$, $V(7)O_5$, and $V(8)O_4$, respectively. In the *ab* plane, units of edgesharing V(1)O₆-V(2)O₆-V(3)O₅ are connected to each other by sharing O(3) and O(5) vertices and are also joined through $V(4)O_4$ by sharing O(1), O(4), and O(6) vertices, forming a V₂O₅ sheet. Two V₂O₅ sheets are connected by sharing O(3)-O(5) edges between $V(1)O_6$ and $V(2)O_6$, making up a V_2O_5 layer. It should be useful to describe the structure of the V_2O_5 layer from another standpoint by utilizing an atomic sheet model as depicted in Fig. 3. A V_2O_5 sheet is composed of $VO_{1.5}$ and O sheets; V and O(1)-O(6) atoms at z = 0.08-0.16 are condensed into a



FIG. 2. Structure of V_2O_5 layer of σ -Zn_{0.25} $V_2O_5 \cdot H_2O$ projected onto the *ab* plane.

 $VO_{1.5}$ sheet and O(13)-O(16) atoms at z = 0.28-0.31 into an O sheet. Two face-to-face V₂O₅ sheets are joined by V(1)-O(3) (actually V(1)-O(9) and V(5)-O(3) in Table 3) and V(2)–O(5) (actually V(2)–O(11) and V(6)–O(5) in Table 3) bonding to form a V_2O_5 layer. Consequently, the V_2O_5 layer of a σ phase is regarded as a double-sheet type being analogous to the V₂O₅ layer of δ phase (12). It is interesting to make a comparison of V₂O₅ layer structures between σ and δ phases as visualized in Fig. 4 for idealized ab planes of V₂O₅ sheets. The arrangement of V–O polyhedra of δ phase is that zigzag chains of edge-sharing VO₆ octahedra running along the b axis are connected by vertex sharing (Fig. 4b) while that of σ phase is that edge-sharing triads and tetrahedra are connected by vertex sharing (Fig. 4a). As expected from the structural relation of the ab plane between both phases, the area of the ab plane for σ phase (85.64 Å²) is close to twice that of δ phase (42.6– 43.8 Å²).

An interlayer Zn atom is octahedrally coordinated, as depicted in Fig. 5, by two apical oxygens of $V(4)O_4$ on opposite sides and by four coplanar water molecules with bond distances and angles listed in Table 4. Since Z(1) and Z(2) are located in equivalent positions of $P2_1/a$, Zn(1)O₆ and Zn(2)O₆ octahedra are reduced to one ZnO₆ of Zn- $(O(16) \times 2, Ow(1) \times 2, Ow(2) \times 2)$ under $P2_1/a$. It should be noted that Zn atom is placed in the position right above and below the apical O(16) atoms, which seems to determine the relative locations of adjacent V₂O₅ layers. The apical O(16) atom is attracted by an Zn atom, resulting in a larger z parameter (z = 0.31) than those (z = 0.28 – 29) of other apical oxygens. Concomitantly the tetrahedral V(4) atom also exhibits a z parameter (z = 0.16) larger than those (z = 0.14) of other V atoms. The values of x and *n* in $Zn_xV_2O_5 \cdot nH_2O$ must not exceed 0.25 and 1.0, respectively, yielding a limited composition of $Zn_{0.25}V_2$ $O_5 \cdot H_2O$.

	V(1)O ₆ o	ctahedron	
$\begin{array}{lll} V(1)-O(1) & 2.00(2) \\ V(1)-O(6) & 2.03(2) \\ O(1)-V(1)-O(2) \\ O(1)-V(1)-O(6) \\ O(1)-V(1)-O(9)^a \\ O(2)-V(1)-O(6) \\ O(2)-V(1)-O(9)^a \\ O(5)-V(1)-O(13) \\ O(6)-V(1)-O(13) \\ O(13)-V(1)-O(9)^a \end{array}$	$\begin{array}{c} V(1)-O(2)\\ V(1)-O(13)\\ 78.0(7)\\ 159.7(7)\\ 78.6(7)\\ 91.8(7)\\ 78.9(7)\\ 104.0(9)\\ 100.6(9)\\ 175.6(9) \end{array}$	$\begin{array}{cccc} 1.94(2) & V(1)-O(5) \\ 1.59(2) & V(1)-O(9)^a \\ O(1)-V(1)-O(5) \\ O(1)-V(1)-O(13) \\ O(2)-V(1)-O(5) \\ O(2)-V(1)-O(6) \\ O(5)-V(1)-O(6) \\ O(5)-V(1)-O(9)^a \\ O(6)-V(1)-O(9)^a \end{array}$	$\begin{array}{c} 1.92(2)\\ 2.37(2)\\ 91.7(7)\\ 98.8(9)\\ 151.2(7)\\ 104.2(9)\\ 89.0(7)\\ 72.7(7)\\ 82.3(6)\end{array}$
	$V(2)O_{6}$ o	ctahedron	
$\begin{array}{lll} V(2)-O(1) & 1.99(2) \\ V(2)-O(4) & 1.90(1) \\ O(1)-V(2)-O(2) \\ O(1)-V(2)-O(4) \\ O(1)-V(2)-O(1)^a \\ O(2)-V(2)-O(4) \\ O(2)-V(2)-O(1)^a \\ O(3)-V(2)-O(14) \\ O(4)-V(2)-O(11)^a \end{array}$	$\begin{array}{c} V(2)-O(2)\\ V(2)-O(14)\\ 77.1(7)\\ 155.4(7)\\ 83.2(6)\\ 79.4(7)\\ 73.6(7)\\ 105(1)\\ 96.1(9)\\ 178.2(9) \end{array}$	$\begin{array}{ccc} 1.99(2) & V(2)-O(3) \\ 1.54(2) & V(2)-O(11)^a \\ O(1)-V(2)-O(3) \\ O(1)-V(2)-O(14) \\ O(2)-V(2)-O(3) \\ O(2)-V(2)-O(14) \\ O(3)-V(2)-O(14) \\ O(3)-V(2)-O(11)^a \\ O(4)-V(2)-O(11)^a \end{array}$	$\begin{array}{c} 1.74(2)\\ 2.50(2)\\ 91.5(8)\\ 98.0(9)\\ 146.5(8)\\ 108(1)\\ 104.3(7)\\ 73.8(7)\\ 83.2(6)\end{array}$
	V(3)O ₅ trigo	nal bipyramid	
$\begin{array}{l} V(3)-O(3) & 1.92(2) \\ V(3)-O(10)^d & 2.14(2) \\ O(3)-V(3)-O(5)^b \\ O(3)-V(3)-O(10)^d \\ O(5)^b-V(3)-O(8)^c \\ O(5)^b-V(3)-O(15) \\ O(8)^c-V(3)-O(15) \\ \end{array}$	V(3)-O(5) ^b V(3)-O(15) 93.2(8) 79.1(7) 72.1(7) 105.0(9) 175.9(8) V(4)O: It	$\begin{array}{cccc} 1.75(2) & V(3)-O(8)^c\\ 1.60(2) & \\ O(3)-V(3)-O(8)^c\\ O(3)-V(3)-O(15) & \\ O(5)^b-V(3)-O(10)^d\\ O(8)^c-V(3)-O(10)^d\\ O(10)^d-V(3)-O(15) & \\ \end{array}$	$\begin{array}{c} 1.91(2)\\ 72.4(7)\\ 105.3(9)\\ 156.3(8)\\ 84.3(6)\\ 98.7(8)\end{array}$
	(4)04 k		
$\begin{array}{lll} V(4)-O(4) & 1.80(1) \\ V(4)-O(16) & 1.62(2) \\ O(4)-V(4)-O(6)^{b} \\ O(4)-V(4)-O(16) \\ O(6)^{b}-V(4)-O(16) \end{array}$	V(4)-O(6)" 109.2(8) 106.3(8) 109.2(9) V(5)O(40)	$\begin{array}{ccc} 1.68(2) & V(4)-O(7)^{\mu} \\ O(4)-V(4)-O(7)^{\mu} \\ O(6)^{\mu}-V(4)-O(7)^{\mu} \\ O(7)^{\mu}-V(4)-O(16) \end{array}$	1.82(2) 115.5(7) 110.3(8) 106.2(9)
	1(5)080		1.0.(2)
$\begin{array}{lll} V(5)-O(1) & 1.99(2) \\ V(5)-O(12) & 2.03(2) \\ O(7)-V(5)-O(8) \\ O(7)-V(5)-O(12) \\ O(7)-V(5)-O(3)^a \\ O(8)-V(5)-O(12) \\ O(8)-V(5)-O(3)^a \\ O(11)-V(5)-O(17) \\ O(12)-V(5)-O(17) \\ O(12)-V(5)-O(3)^a \end{array}$	V(5)-O(8) $V(5)-O(17)$ $76.5(7)$ $160.2(7)$ $91.9(7)$ $77.3(7)$ $102.6(9)$ $99.0(9)$ $176.0(9)$	$\begin{array}{l} 1.95(2) & V(5)-O(11) \\ 1.62(2) & V(5)-O(3)^{a} \\ O(7)-V(5)-O(11) \\ O(7)-V(5)-O(17) \\ O(8)-V(5)-O(17) \\ O(8)-V(5)-O(17) \\ O(11)-V(5)-O(12) \\ O(11)-V(5)-O(3)^{a} \\ O(12)-V(5)-O(3)^{a} \end{array}$	$\begin{array}{c} 1.94(2)\\ 2.39(2)\\ 92.9(7)\\ 99.6(9)\\ 150.6(7)\\ 106(1)\\ 89.5(7)\\ 73.8(7)\\ 82.9(6)\end{array}$
	V(6)O ₆ o	ctahedron	
$\begin{array}{l} V(6)-O(7)^b & 1.94(2) \\ V(6)-O(10) & 1.92(2) \\ O(7)^b-V(6)-O(8)^b \\ O(7)^b-V(6)-O(10) \\ O(7)^b-V(6)-O(10) \\ O(8)^b-V(6)-O(10) \\ O(8)^b-V(6)-O(10) \\ O(8)^b-V(6)-O(18) \\ O(10)-V(6)-O(18) \\ O(18)-V(6)-O(5)^a \end{array}$	V(6)-O(8) ^b V(6)-O(18) 77.2(7) 155.2(7) 83.6(7) 79.1(7) 72.7(7) 104(1) 97.3(9) 177.5(8)	$\begin{array}{cccc} 1.98(2) & V(6)-O(9) \\ 1.57(2) & V(6)-O(5)^a \\ O(7)^b-V(6)-O(18) \\ O(8)^b-V(6)-O(18) \\ O(8)^b-V(6)-O(18) \\ O(9)^b-V(6)-O(118) \\ O(9)-V(6)-O(10) \\ O(9)-V(6)-O(5)^a \\ O(10)-V(6)-O(5)^a \end{array}$	$\begin{array}{c} 1.74(2)\\ 2.46(2)\\ 92.6(8)\\ 97.5(9)\\ 145.3(8)\\ 109.6(9)\\ 102.9(8)\\ 73.3(7)\\ 82.5(6)\end{array}$
	V(7)O ₅ trigo	nal bipyramid	
$\begin{array}{lll} V(7)-O(9) & 1.93(2) \\ V(7)-O(4) & 2.14(2) \\ O(9)-V(7)-O(1)^d \\ O(9)-V(7)-O(4) \\ O(11)-V(7)-O(2) \\ O(11)-V(7)-O(19) \\ O(2)-V(7)-O(19) \\ \end{array}$	$ \begin{array}{c} V(7)-O(11)^d \\ V(7)-O(19) \\ 92.5(7) \\ 79.7(6) \\ 72.2(7) \\ 105.6(9) \\ 176.0(8) \end{array} $	$\begin{array}{ccc} 1.73(2) & V(7)-O(2) \\ 1.58(2) & \\ O(9)-V(7)-O(2) & \\ O(9)-V(7)-O(19) & \\ O(11)-V(7)-O(4) & \\ O(2)-V(7)-O(4) & \\ O(4)-V(7)-O(19) & \\ \end{array}$	1.89(2) 71.1(6) 105.9(9) 156.9(7) 84.6(6) 97.5(8)
	$V(8)O_4$ te	etrahedron	
$\begin{array}{ll} V(8)-O(10)^d & 1.80(2) \\ V(8)-O(20) & 1.60(2) \\ O(10)^d-V(8)-O(12)^d \\ O(10)^d-V(8)-O(20) \\ O(12)^d-V(8)-O(20) \\ \end{array}$	$V(8)-O(12)^d$ 109.4(8) 107.0(9) 107.5(9)	1.66(2) V(8)-O(1) $O(10)^d$ -V(8)-O(1) $O(12)^d$ -V(8)-O(1) O(1)-V(8)-O(20)	1.79(2) 116.7(7) 109.8(8) 105.9(9)

Note. Symmetry codes: ${}^{a}1 - x, 1 - y, -z; {}^{b}x, y - 1, z; {}^{c}x, y - 1, 1 + z; {}^{d}1 + x, y, z.$



FIG. 3. Representation for double-sheet-type V_2O_5 layers for σ -Zn_{0.25} $V_2O_5 \cdot H_2O$ viewed along the *b* axis.



FIG. 4. Polyhedral representations for idealized structures of V_2O_5 sheet in the *ab* plane for (a) σ phase and (b) δ phase.



FIG. 5. Locations of interstitial Zn atoms and water molecules in a interlayer region between V_2O_5 layers. Zn atoms and water molecules are denoted by small and large circles, respectively, and a V_2O_5 layer is indicated by a polyhedral representation of a V_2O_5 sheet.

Lattice Distortion into Triclinic System

The structure of σ -phase structure is well described based on the monoclinic $P2_1/a$ but actually adopts the triclinic system. As described above, the positions of V and O atoms in the V_2O_5 layer as well as interlayer Zn atoms show the approximate relation of (x, y, z)-(x + 1/2, 1/2 - y, z) derived from the symmetry operation of $P2_1/a$. However as can be seen from Table 4, the relation is not applied to the positions of interlayer water molecules. Instead, the approximate relation of (x, y, z)-(x + 1/2,y + 1/2, z) is found for Ow(1)–Ow(3) and Ow(2)–Ow(4), which corresponds to the C-type lattice or consistent with the positional relation of Zn atoms. Therefore interlayer water molecules possibly cause the lattice distortion from monoclinic to triclinic. Figure 6 visualizes the locations of water molecules for the $P2_1/a$ symmetry in comparison with the Ctype symmetry, where Ow(1)-Ow(3') and Ow(2)-Ow(4')have the (x, y, z) - (x + 1/2, 1/2 - y, z) of $P2_1/a$. It is quite

likely that $P2_1/a$ is realized when Ow(3) and Ow(4) are replaced by Ow(3') and Ow(4'), respectively.

Phase Change into Anhydrous σ Phase

Hydrated σ -Zn_{0.25}V₂O₅ · H₂O was converted into anhydrous σ -Zn_{0.25}V₂O₅ by heating as demonstrated by the TG-DTA diagram in Fig. 7. At a heating rate of 5°C min⁻¹ dehydration gradually proceeded above 70°C, developed rapidly around 170°C, and finished around 200°C accompanied by a broad endothermic DTA peak. The anhydrous phase was examined by high-temperature X-ray diffraction, and the pattern at 200°C is presented together with the pattern at RT in Fig. 8. The layer spacing was found to contract from 10.76 to 8.92 Å by dehydration. In the anhydrous phase Zn atoms must shift to an appropriate site which is coordinated by more apical oxygens than in the hydrated phase. The structural details of the anhydrous phase, however, remain unknown.

	$Zn(1)O_6$ of	ctahedron	
$Zn(1)$ – $Ow(1)^{a,b}$	2.11(2)	$Zn(1)$ – $Ow(2)^{c,d}$	2.13(2)
$Zn(1)-O(20)^{c,d}$	2.06(2)		
$Ow(1)^a - Zn(1)Ow(1)^b$	180.0(1)	$Ow(1)^{a}-Zn(1)-Ow(2)^{c}$	88.8(8)
$Ow(1)^a - Zn(1)Ow(2)^b$	91.2(8)	$Ow(1)^{a}-Zn(1)-O(20)^{c}$	90.9(8)
$Ow(1)^{a}-Zn(1)O(20)^{d}$	89.1(8)	$Ow(1)^{b}-Zn(1)-Ow(2)^{c}$	91.2(8)
$Ow(1)^b - Zn(1)Ow(2)^d$	88.8(8)	$Ow(1)^b - Zn(1) - O(20)^c$	89.1(8)
$Ow(1)^{b}-Zn(1)O(20)^{d}$	90.9(8)	$Ow(2)^c - Zn(1) - Ow(2)^d$	180.0(1)
$Ow(2)^{c} - Zn(1)O(20)^{c}$	89.0(9)	$Ow(2)^{c}-Zn(1)-O(20)^{d}$	89.0(9)
$Ow(2)^d - Zn(1)O(20)^c$	89.0(9)	$Ow(2)^d - Zn(1) - O(20)^d$	91.0(9)
$O(20)^{c} - Zn(1)O(20)^{d}$	180.0(1)		
	$Zn(2)O_6$ of	ctahedron	
$Zn(2)$ – $Ow(3)^{a,e}$	2.12(2)	$Zn(2)$ – $Ow(4)^{a,e}$	2.11(2)
$Zn(2)-O(16)^{a,e}$	2.03(2)		
$Ow(3)^a - Zn(2)Ow(3)^e$	180.0(1)	$Ow(3)^{a}-Zn(2)-Ow(4)^{e}$	89.6(8)
$Ow(3)^a - Zn(2)Ow(4)^e$	90.4(8)	$Ow(3)^{a}-Zn(2)-O(16)^{a}$	93.0(9)
$Ow(3)^{a} - Zn(2)O(16)^{e}$	87.0(9)	$Ow(3)^e$ - $Zn(2)$ - $Ow(4)^a$	90.4(8)
$Ow(3)^e - Zn(2)Ow(4)^e$	89.6(8)	$Ow(3)^{e}-Zn(2)-O(16)^{a}$	87.0(9)
$Ow(3)^{e} - Zn(2)O(16)^{e}$	93.0(9)	$Ow(4)^{a}-Zn(2)-Ow(4)^{e}$	180.0(1)
$Ow(4)^{a} - Zn(2)O(16)^{a}$	90.9(8)	$Ow(4)^{a}-Zn(2)-O(16)^{e}$	89.1(8)
$Ow(4)^{e} - Zn(2)O(16)^{a}$	89.1(8)	$Ow(4)^{e} - Zn(2) - O(16)^{e}$	90.9(8)
	100.0(1)		

 TABLE 4

 Bond Distances (Å) and Angles (°) for Zn–O Octahedra



Formation of δ -Phase Zn Compounds

A δ -phase Zn compound was easily obtained from other δ -phase compounds such as a vanadyl intercalate



FIG. 6. Positional shifts of Ow3 and Ow4 to Ow3' and Ow4', respectively, to meet the $P_{2_1/a}$ symmetry. Ow3' and Ow4' are obtained from Ow1 and Ow2, respectively, by the translation (x + 1/2, 1/2 - y, z).



FIG. 7. TG-DTA diagram for σ -Zn_{0.25}V₂O₅ · H₂O recorded at a heating rate of 5°C min⁻¹.

 $(VO)_{0.1}V_2O_5 \cdot nH_2O$ or potassium intercalate $K_{0.3}V_2O_5 \cdot$ nH_2O by ion-exchange treatment of soaking in a ZnCl₂ solution. This δ -phase Zn compound was formulated by δ -Zn_{0.16}V₂O₅ · *n*H₂O and found to exhibit reversible transitions between the phases with different layer spacings and degrees of hydration: 14.15 Å in wet (14.2-Å phase) 13.48 Å ($n \approx 2.2$) at RT (13.5-Å phase), 10.43 Å ($n \approx 1.0$) at 70°C (10.4-Å phase), and 9.31 Å (n = 0) at 180°C (9.3-Å phase). The same δ -phase Zn compound exhibiting the 13.5-Å phase at RT was directly synthesized by the hydrothermal treatment of VOSO₄-ZnSO₄ mixed solutions at 220 to 280°C. The present hydrothermal system of ZnCl₂ VO(OH)₂ also produced a δ -phase compound when reaction temperatures below 250°C were employed. This δ phase, which exhibited the 10.4-Å phase at RT instead of the 13.5-Å phase, was found to have higher Zn content as formulated by δ -Zn_{0.25}V₂O₅ · nH₂O (n \approx 1). The δ -phase Zn compounds may be compared with σ -phase analogues; the 10.4-Å δ phase corresponds to the hydrated 10.75-Å σ phase and the anhydrous 9.3-Å δ phase to the anhydrous 8.98-Å σ phase. It is noted that a difference between δ and σ phase Zn compounds was also found in the ionexchange activity that δ phase showed high ion-exchange activity while σ phase showed no or little activity.



FIG. 8. Powder X-ray diffraction patterns for σ -Zn_{0.25}V₂O₅·H₂O taken at RT (bottom) and 200°C (top).

CONCLUDING REMARKS

A new layered phase of vanadium bronze designated as σ -Zn_{0.25}V₂O₅ · H₂O has been hydrothermally synthesized. A single-crystal study revealed a novel structure of a double-sheet-type V₂O₅ layer as well as interstitial sites of Zn atom and water molecule. The present study adds a new member to the family of vanadium bronzes having V₂O₅ layers. This new member appears to be related to the existing member of δ phase in terms of a double-sheet-type structure. The structure of σ phase is featured by VO₄ tetrahedra coexisting with VO₆ and VO₅ polyhedra while that of δ phase is made up of VO₆ octahedra. Interstitial Zn atom is octahedrally coordinated by two apical oxygens

of VO₄ tetrahedra on opposite sides and by four water molecules in a midplane. In the anhydrous phase this coordination must be drastically changed to contract the layer spacing by 1.83 Å. Production of isomorphous σ phases has been observed for other divalent *M* ions such as M =Mg, Mn, Ni, and Co by our preliminary work, and the details will be reported elsewhere. It seems that σ phase is formed with divalent metal ions while δ phase is formed with both mono- and divalent cations.

ACKNOWLEDGMENTS

We are grateful to Professor S. Sasaki of Tokyo Institute of Technology for preparing and revising the computer programs. The present work is supported by Grant-in-Aid for Scientific Research from the Ministry of Education, Science, Sports and Culture of Japan. The drawings of crystal structures were produced with ATOMS by Shape Software.

REFERENCES

- 1. P. Hagenmuller, Prog. Solid State Chem. 5, 71 (1971).
- 2. K. Waltersson, "Chem. Commun. Univ. Stockholm No. 7" (1976).
- 3. H. G. Bachmann, F. R. Ahmed, and W. H. Barnes, *Z. Kristallogr.* **115**, 110 (1961).
- F. Théobald, R. Cabara, and J. Bernard, J. Solid State Chem. 17, 431 (1976).
- 5. T. Yao, Y. Oka, and N. Yamamoto, Mater. Res. Bull. 27, 669 (1992).
- P. Hagenmuller, J. Galy, M. Pouchard, and A. Cassalot, *Mater. Res.*, *Bull.* 1, 45 (1966).
- 7. P. Hagenmuller, J. Galy, M. Pouchard, and A. Cassalot, *Mater. Res., Bull.* **1**, 95 (1966).
- J. Galy, A. Cassalot, M. Pouchard, and P. Hagenmuller, C. R. Acad. Sci. 262, 1055 (1966).
- 9. S. Andersson, Acta Chem. Scand. 19, 1371 (1965).
- 10. J. Galy, J. Solid State Chem. 100, 229 (1992).
- 11. T. Yao, Y. Oka, and N. Yamamoto, J. Mater. Chem. 2, 331 (1992).
- 12. Y. Oka, T. Yao, and N. Yamamoto, J. Mater. Chem. 5, 1423 (1995).
- 13. T. Yao, Y. Oka, and N. Yamamoto, J. Mater. Chem. 2, 337 (1992).
- 14. J. Galy, D. Lavaud, A. Cassarot, and P. Hagenmuller, J. Solid State Chem. 2, 531 (1970).
- 15. J. Galy, J. Darriet, and P. Hagenmuller, *Rev. Chim. Miner.* **8**, 509 (1971).
- 16. S. Sasaki, private communication.
- 17. S. Sasaki, KEK Internal Report, Vol. 3, National Laboratory of High Energy Physics, Tukuba, Japan, 1987.
- "International Tables for X-ray Crystallography IV." Kynoch Press, Birmingham, UK, 1974.
- TEXSAN: Crystal Structure Analysis Package, Molecular Structure Corp., The Woodland, TX, 1985 & 1992.